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# Predicted superconductivity of Ni<sub>2</sub>VAl and pressure dependence of superconductivity in Ni<sub>2</sub>NbX (X = Al, Ga and Sn) and Ni<sub>2</sub>VAl

P V Sreenivasa Reddy<sup>1</sup>, V Kanchana<sup>1</sup>, G Vaitheeswaran<sup>2</sup> and David J Singh<sup>3</sup>

<sup>1</sup> Department of Physics, Indian Institute of Technology Hyderabad, Kandi, Medak-502 285, Telangana, India

<sup>2</sup> Advanced Centre of Research in High Energy Materials (ACRHEM), University of Hyderabad, Prof C R Rao Road, Gachibowli, Hyderabad 500 046, Telangana, India

<sup>3</sup> Department of Physics and Astronomy, University of Missouri-Columbia, Columbia, MO 65211, USA

E-mail: [kanchana@iith.ac.in](mailto:kanchana@iith.ac.in)

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## Abstract

A first-principles study of the electronic and superconducting properties of the Ni<sub>2</sub>VAl Heusler compound is presented. The electron–phonon coupling constant of  $\lambda_{\text{ep}} = 0.68$  is obtained, which leads to a superconducting transition temperature of  $T_c = \sim 4$  K (assuming a Coulomb pseudopotential  $\mu^* = 0.13$ ), which is a relatively high transition temperature for Ni based Heusler alloys. The electronic density of states reveals a significant hybridization between Ni-*eg* and V-*t<sub>2g</sub>* states around the Fermi level. The Fermi surface, consisting of two electron pockets around the X-points of the Brillouin zone, exhibits nesting and leads to a Kohn anomaly of the phonon dispersion relation for the transverse acoustic mode TA2 along the (1, 1, 0) direction, which is furthermore found to soften with pressure. As a consequence,  $T_c$  and  $\lambda_{\text{ep}}$  vary non-monotonically under pressure. The calculations are compared to similar calculations performed for the Ni<sub>2</sub>NbX (X = Al, Ga and Sn) Heusler alloys, which experimentally have been identified as superconductors. The experimental trend in  $T_c$  is well reproduced, and reasonable quantitative agreement is obtained. The calculated  $T_c$  of Ni<sub>2</sub>VAl is larger than either calculated or observed  $T_c$ s of any of the Nb compounds. The Fermi surfaces of Ni<sub>2</sub>NbAl and Ni<sub>2</sub>NbGa consist of only a single electron pocket around the X point, however under compression a second electron pocket similar to that of Ni<sub>2</sub>VAl emerges Ni<sub>2</sub>NbAl and the  $T_c$  increases non-monotonically in all the compounds. Fermi surface nesting and associated Kohn anomalies are a common feature of all four compounds, albeit weakest in Ni<sub>2</sub>VAl.

Keywords: Fermi surface nesting, charge density wave, Kohn anomaly, superconductivity, pressure effect on superconductivity

(Some figures may appear in colour only in the online journal)

## 1. Introduction

Heusler alloys are ternary intermetallic compounds of the form X<sub>2</sub>YZ, where X is generally a transition metal, Y is yet another transition metal from group VIII B-IB and Z is an *sp* metal or a metalloid. These compounds display a wide range of physical properties including half-metallicity [1–4],

magnetic ordering [5, 6], heavy fermion behaviour [7–9], shape memory effect [10, 11] and thermoelectricity [12, 13]. Some are superconductors having a superconducting transition temperature ( $T_c$ ) ranging from 0.74 K (HfNi<sub>2</sub>Al) to 4.7 K (YPd<sub>2</sub>Sn). Superconductivity was first reported for the Heusler alloys by Ishikawa *et al* [14], where they focused mainly on the systems RPd<sub>2</sub>Sn and RPd<sub>2</sub>Pb with R being a

rare-earth metal. Among the known Heusler superconductors, Ni-based alloys  $\text{Ni}_2\text{NbX}$  ( $X = \text{Al, Ga and Sn}$ ) have attracted much attention due to their intermediate electron phonon coupling constant [15–17]. Interestingly, the  $\text{Ni}_2\text{NbX}$  ( $X = \text{Al, Ga and Sn}$ ) compounds are paramagnets and are found to be superconductors although Ni is a ferromagnet and Ni compounds are often magnetic. Here we predict  $\text{Ni}_2\text{VAl}$  to be a superconductor. Importantly, this provides an experimentally testable prediction that if confirmed and taken in conjunction with the correct predictions for the related compounds, would strongly restrict the possible role of spin-fluctuations associated with Ni magnetism in the superconducting properties of these phases.

In these compounds the presence of Nb works against magnetism associated with Ni leading to the paramagnetic ground states of these compounds [15, 17]. The same paramagnetic nature is recently observed in  $\text{ScT}_2\text{Al}$  ( $T = \text{Ni, Pd, Pt, Cu, Ag, Au}$ ) [18]. The  $\text{Ni}_2\text{NbX}$  superconductors have  $T_c$  of 2.15 K ( $\text{Ni}_2\text{NbAl}$ ) [15], 1.54 K ( $\text{Ni}_2\text{NbGa}$ ) [15] and 2.9 K [15] (3.4 K [16, 17]) ( $\text{Ni}_2\text{NbSn}$ ) with calculated electron phonon coupling constants [15] ( $\lambda_{\text{ep}}$ ) of 0.52, 0.50 and 0.61 respectively. Electronic structures and cohesive properties of  $\text{Ni}_2\text{NbAl}$  and  $\text{Ni}_2\text{VAl}$  were studied by Lin *et al* [19] and a large value of cohesive energy is observed with a pronounced hybridization between Ni-Nb/V-Al atoms. The interaction between these atoms creates deep valleys in density of states which separate bonding and anti-bonding region. It is this type of covalency that works against magnetism.

In prior work [20], we have studied electronic structure of  $\text{Ni}_2\text{NbAl}$  and  $\text{Ni}_2\text{VAl}$  and superconductivity of  $\text{Ni}_2\text{NbAl}$  both at ambient as well as under compression and observed a change in FS topology at a pressure of around 17 GPa in  $\text{Ni}_2\text{NbAl}$ . The change in the FS topology leads to the non-monotonic variation in the superconducting transition temperature. In case of  $\text{Ni}_2\text{VAl}$  we did not find any FS topology changes in the pressure range studied. Here we focus in detail on the superconductivity of  $\text{Ni}_2\text{VAl}$  and report results on  $\text{Ni}_2\text{NbGa}$  and  $\text{Ni}_2\text{NbSn}$ . No other reports are available for  $\text{Ni}_2\text{NbGa}$  and  $\text{Ni}_2\text{NbSn}$  regarding the electronic structure, elastic and vibrational properties.

Superconductivity in conventional intermetallics has been traditionally discussed in terms of the density of states at the Fermi level and the number of valence electrons per atom. From the Matthias rule [21, 22], the number of valence electrons per atom should be close to five or seven. Even though the Heusler superconducting compounds follow the above prescription, the superconducting transition temperatures of these compounds are relatively low. Among the compositional elements of the  $\text{Ni}_2\text{VAl}$ , vanadium is reported to be a superconductor with superconducting transition temperature ( $T_c$ ) of 5.3 K [23] and 3.6 K [24]. From previous literature [24], the total density of states (DOS) value of bcc vanadium is 1.46 (states  $\text{eV}^{-1}\text{f.u.}^{-1}$ ). Pressure effect on the  $T_c$  has been studied experimentally and theoretically for bcc vanadium [23–26] upto 120 GPa in which the  $T_c$  is found to increase linearly with pressure. However, high DOS also favours spin-fluctuations, which work against electron–phonon superconductivity [27]. From our previous work the total DOS of  $\text{Ni}_2\text{VAl}$  is observed

to be more than the value of vanadium and the number of electrons/atom is observed to be seven in the present compound obeying the Matthias rule.

We also studied the pressure dependence, as pressure provides a tuning parameter that is very useful in understanding trends and mechanisms. Previous studies on Hf based Heusler alloys have shown an increase in the superconducting transition temperature with decreasing lattice parameter [28], while certain other cases show an increase in  $T_c$  with increase in lattice constant [28]. We note that the behaviour can also be more complex under pressure [29]. Fundamentally, superconductivity is an instability of the FS, and so pressure induced changes in the FS can lead to non-trivial changes in superconducting properties.

Another interesting feature that connects with superconductivity is the presence of van Hove singularities [30], in the electronic structure, leading to peaks in the density of states as found in some of the Pd based Heusler compounds [31–33]. Interesting behaviour might be anticipated if the Fermi level can be brought to this peak by means of alloying. Another noteworthy point present in these compounds is the softening of the TA2 acoustic phonon modes, which can be well correlated with the FS nesting and the corresponding nesting vector decides the position of the Kohn anomaly present in these compounds. Further to these, we also find a softening in the acoustic phonons under pressure leading to the variation in the  $T_c$ , which is discussed in detail.

The organization of the paper is as follows. Computational details are presented in the section 2. Results and discussions of ground state, electronic structure, mechanical, vibrational, superconducting properties and pressure effect on these properties are presented in the section 3. The conclusions are given in section 4.

## 2. Computational details

The full potential linearized augmented plane wave (FP-LAPW) method as implemented in Wien2k code [34] is used to calculate the ground state and electronic structure of the present compounds. We adopt the generalized gradient approximation (GGA) of Perdew–Burke–Ernzerhof (PBE) [35]. The wave functions are expanded up to angular momentum  $l = 10$  inside the muffin-tin spheres. The radii of muffin tin spheres  $R_{\text{MT}}$  were 1.78 a.u for Ni, Nb and V, 1.67 a.u for Al, 2.0 a.u for Ga and 2.3 a.u for Sn atoms. The plane wave cut off energy is used  $K_{\text{max}} = 9/R_{\text{MT}}$ , where  $R_{\text{MT}}$  is the smallest muffin-tin radius and  $K_{\text{max}}$  is the magnitude of largest plane wave expansion. All the electronic structure calculations are performed with a  $44 \times 44 \times 44$   $k$ -mesh in the Monkhorst–Pack [36] scheme which gives 2168  $k$ -points in the irreducible part of the Brillouin zone (BZ). Tetrahedron method [37] was used to integrate the BZ. Energy convergence up to  $10^{-5}$  Ry is used to get the proper convergence of the self consistent calculation. The Birch–Murnaghan [38] equation of state is used to find the equilibrium lattice parameter and bulk modulus by fitting the total energies as a function of primitive cell volume. We have not found any significant change in the electronic structure at the Fermi level with the inclusion of

**Table 1.** Ground state properties of Ni<sub>2</sub>NbX (X = Al, Ga, Sn) and Ni<sub>2</sub>VAl at ambient pressure combined with experimental reports.

Parameters	Ni <sub>2</sub> NbAl	Ni <sub>2</sub> NbGa	Ni <sub>2</sub> NbSn	Ni <sub>2</sub> VAl
$a_{\text{exp}}$ (Å)	5.969 [15], 5.9755 [40]	5.956 [15]	6.157 [15], 6.160 [16], 6.179 [17]	5.8031 [40], 6.33 [41]
$a_{\text{th}}$ (Å)	5.996, 6.00 [19]	5.991	6.202	5.800, 5.78 [19]
$B$ (GPa)	181	182	170	181
$\gamma_{\text{exp}}$ (mJ mol <sup>-1</sup> K <sup>2</sup> )	8.0 [15], 11.03 [40]	6.5 [15]	4.0 [15], 5.15 [16]	14.18 [40]
$\gamma_{\text{th}}$ (mJ mol <sup>-1</sup> K <sup>2</sup> )	5.36	5.19	5.52	8.27

Note: Where  $a_{\text{exp}}$  is an experimental lattice parameter,  $a_{\text{th}}$  is a theoretical lattice parameter,  $B$  is bulk modulus,  $\gamma_{\text{exp}}$  is the experimental Sommerfeld coefficient and  $\gamma_{\text{th}}$  is the theoretical Sommerfeld coefficient.

spin-orbit coupling (SOC). So, the reported calculations are without SOC.

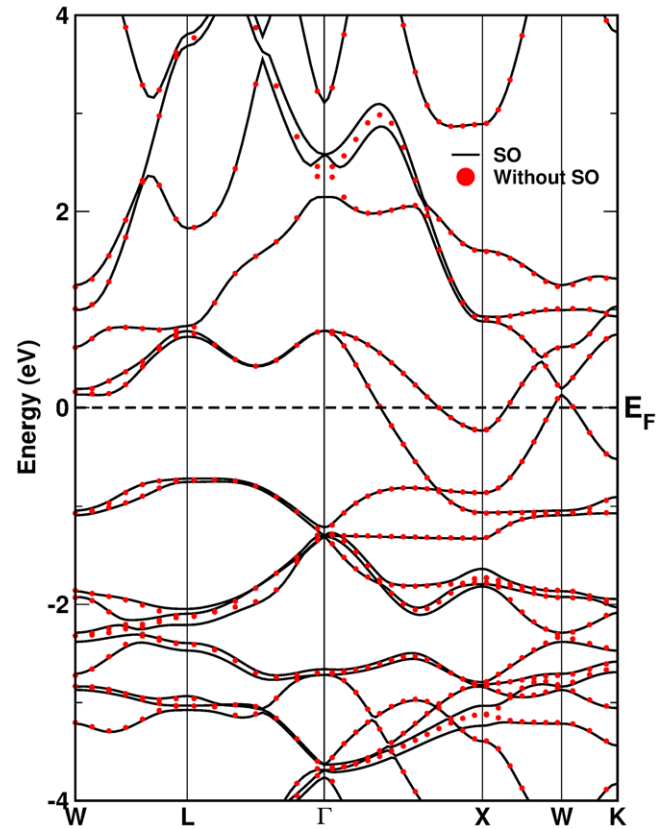
Phonon dispersions and electron-phonon interaction calculations were performed using the plane wave pseudopotential method (PWSCF) which is implemented in QUANTUM ESPRESSO [39] code. The GGA-PBE exchange correlation functional is used in the present calculations for all the compounds. The electron-ion interaction is described by using norm-conserving pseudopotentials. The maximum plane wave cut-off energy ( $ecutwfc$ ) was 90 Ry and the electronic charge density was expanded up to 360 Ry (in the case of Ni<sub>2</sub>VAl it is 140 and 560 Ry). A  $16 \times 16 \times 16$   $k$ -points grid within the BZ is used for the phonon calculations. Gaussian broadening of 0.02 Ry and a  $4 \times 4 \times 4$  uniform grid of  $q$ -points are used for phonon calculations.

### 3. Results and discussions

#### 3.1. Ground state, electronic structure properties

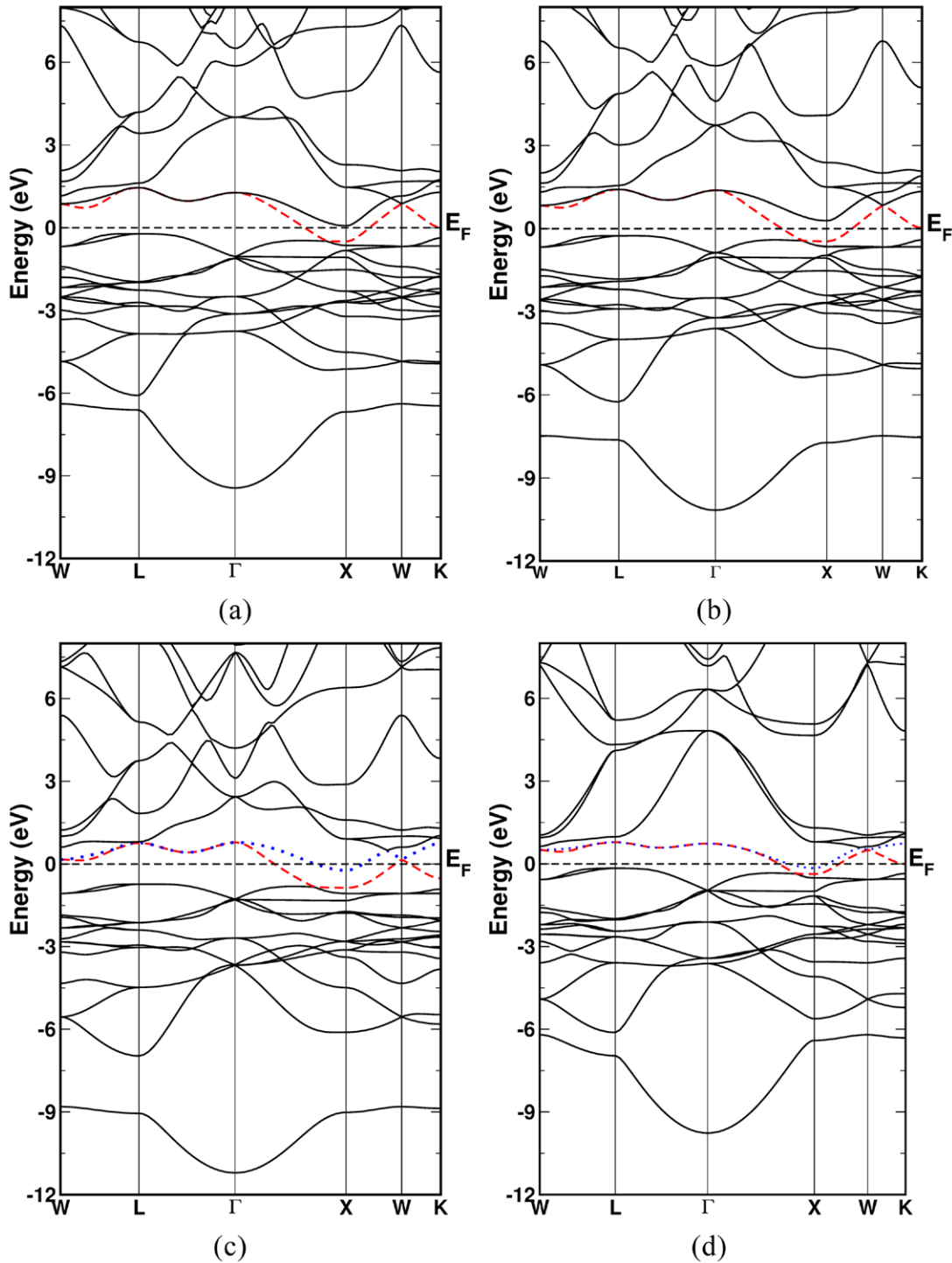
The ground state properties of Ni<sub>2</sub>NbX (X = Al, Ga and Sn) and Ni<sub>2</sub>VAl are evaluated using the experimental lattice parameter and atomic positions [15–17, 40, 41]. The calculated equilibrium lattice constant and bulk modulus values are presented in table 1 along with the available experimental and other theoretical work and a good agreement is seen between the present values and earlier reports. The calculated bulk moduli of Ni<sub>2</sub>NbAl, Ni<sub>2</sub>NbGa and Ni<sub>2</sub>VAl compounds are nearly same and found to be slightly high than Ni<sub>2</sub>NbSn.

We have calculated the band structure along the high symmetry directions in the irreducible BZ with and without inclusion of SOC. From figure 1 (given for only Ni<sub>2</sub>NbSn) it is seen that the SOC effect is very small around the Fermi level and we have proceeded with the further calculations excluding SOC. The band structures for all the compounds are given in figure 2. The overall band shapes are similar for all the compounds studied but the Fermiology is different as seen from the band crossing at  $E_F$ . For all the compounds the lowest lying valence band arises mainly from the  $s$ -states of X (X = Al, Ga, Sn) atom. The bands near the  $E_F$  are due to the hybridization of both Ni and Nb/V atoms. The band which cross the  $E_F$  at X point in both Ni<sub>2</sub>NbAl and Ni<sub>2</sub>NbGa is due to the  $d_{eg}$  states of Ni. The same band in Ni<sub>2</sub>NbSn is crossing the  $E_F$  at both X and W points. In addition to that we have an extra band at the X point in Ni<sub>2</sub>NbSn which has Nb  $d$  character. In the case of Ni<sub>2</sub>VAl we have two bands to cross the  $E_F$  at the X point.



**Figure 1.** Band structure for Ni<sub>2</sub>NbSn with and without inclusion of the spin-orbit effect at the theoretical equilibrium volume.

The electronic DOS is shown in figure 3 along with the atom projected DOS. Even though these compounds are composed of different elements from different rows in the periodic table, the total DOS for all the compounds looks similar reflecting the similar band shapes. For all the compounds we observe valleys at energies of  $-6$  eV,  $-1$  eV,  $0.5$  eV. In the case of Ni<sub>2</sub>NbSn there is another valley at around  $-3$  eV. This feature indicates that the interaction between the constituent atom is strong [19]. Among all the compounds Ni<sub>2</sub>VAl has the highest value of DOS at  $E_F$  with 3.51 states  $eV^{-1}$  and remaining Ni<sub>2</sub>NbX compounds have 2.27, 2.20 and 2.34 states  $eV^{-1}$  for X = Al, Ga and Sn respectively. From the atom projected DOS we have observed that the primary contribution to the total DOS at  $E_F$  is due to Ni atom ( $d_{eg}$  states), the secondary contribution is due to Nb/V atom ( $d_{2g}$  states) and the least contribution arises from X atom ( $p$  states) (X = Al, Ga and Sn). The calculated Sommerfeld coefficient  $\gamma$  is also given in the table 1 along with the



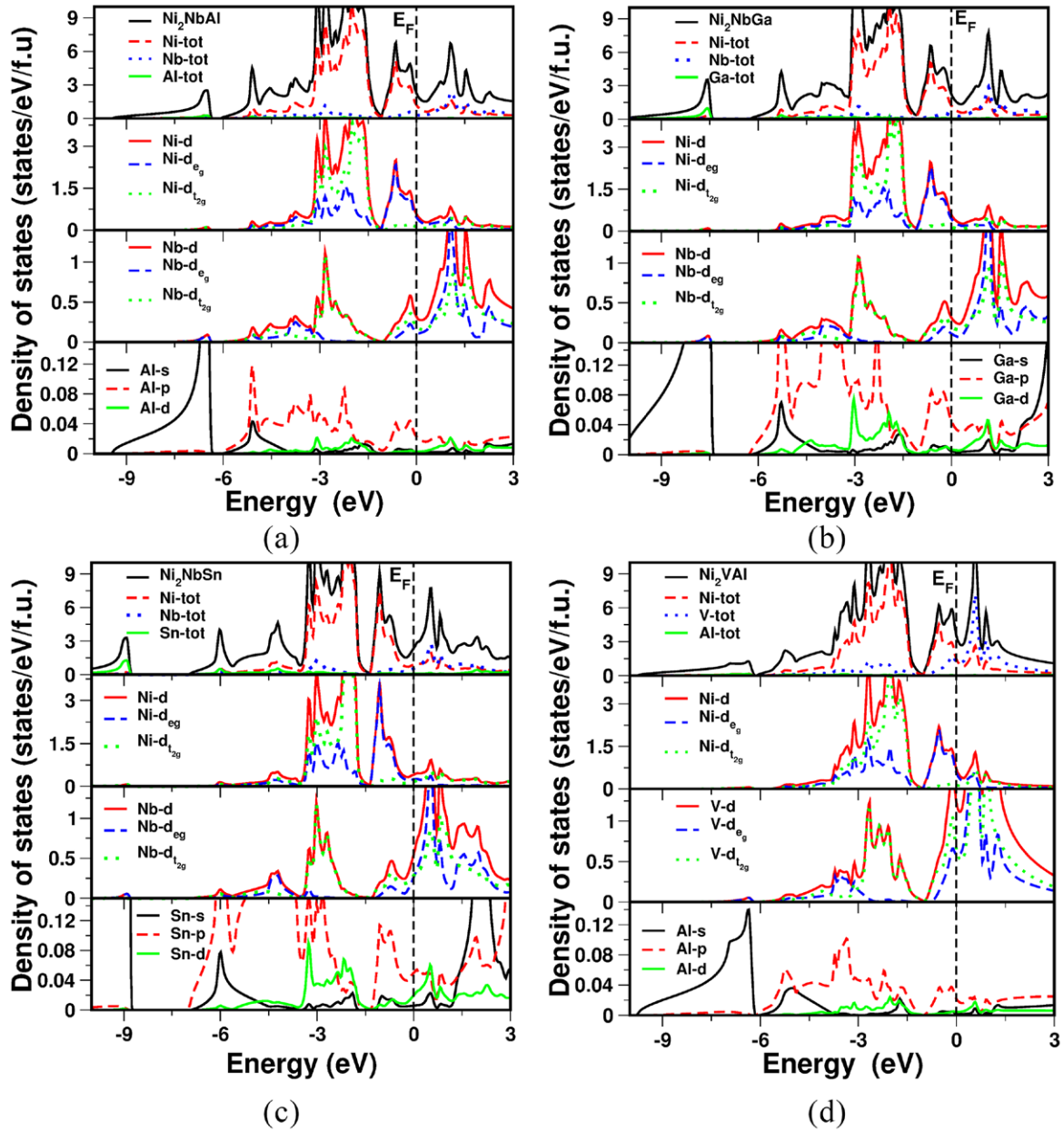
**Figure 2.** Band structure for (a)  $\text{Ni}_2\text{NbAl}$ , (b)  $\text{Ni}_2\text{NbGa}$ , (c)  $\text{Ni}_2\text{NbSn}$  and (d)  $\text{Ni}_2\text{VAl}$ .

experimental values. There is a decreasing trend in  $N(E_F)$  from  $\text{Ni}_2\text{VAl} \rightarrow \text{Ni}_2\text{NbSn} \rightarrow \text{Ni}_2\text{NbAl} \rightarrow \text{Ni}_2\text{NbGa}$ . Experimental specific heat can be calculated with  $(1 + \lambda) \times \gamma_{\text{th}}$ . The  $\lambda$  in the enhancement factor is related to but not identical to the superconducting  $\lambda$  and includes in addition contributions from spin fluctuations and other interactions if present. The inferred values are in the range  $\sim 0.5\text{--}1$  for  $\text{Ni}_2\text{NbAl}$  and  $\text{Ni}_2\text{VAl}$ , in reasonable accord with the calculated superconducting  $\lambda$  below. The values for  $\text{Ni}_2\text{NbGa}$  and especially  $\text{Ni}_2\text{NbSn}$  are

anomalously low. The origin of this is not clear, and warrants further investigation. Site disorder in samples is one possibility. In any case, we also note that the  $N(E_F)$  are not high enough to place any of the compounds near Stoner criterion for ferromagnetism.

The van Hove singularity is observed in both valence and conduction bands at the L-point close to  $E_F$  around 1 eV and  $-1$  eV energy range. From earlier available reports [31–33], one saddle point is observed in Pd based compounds, at





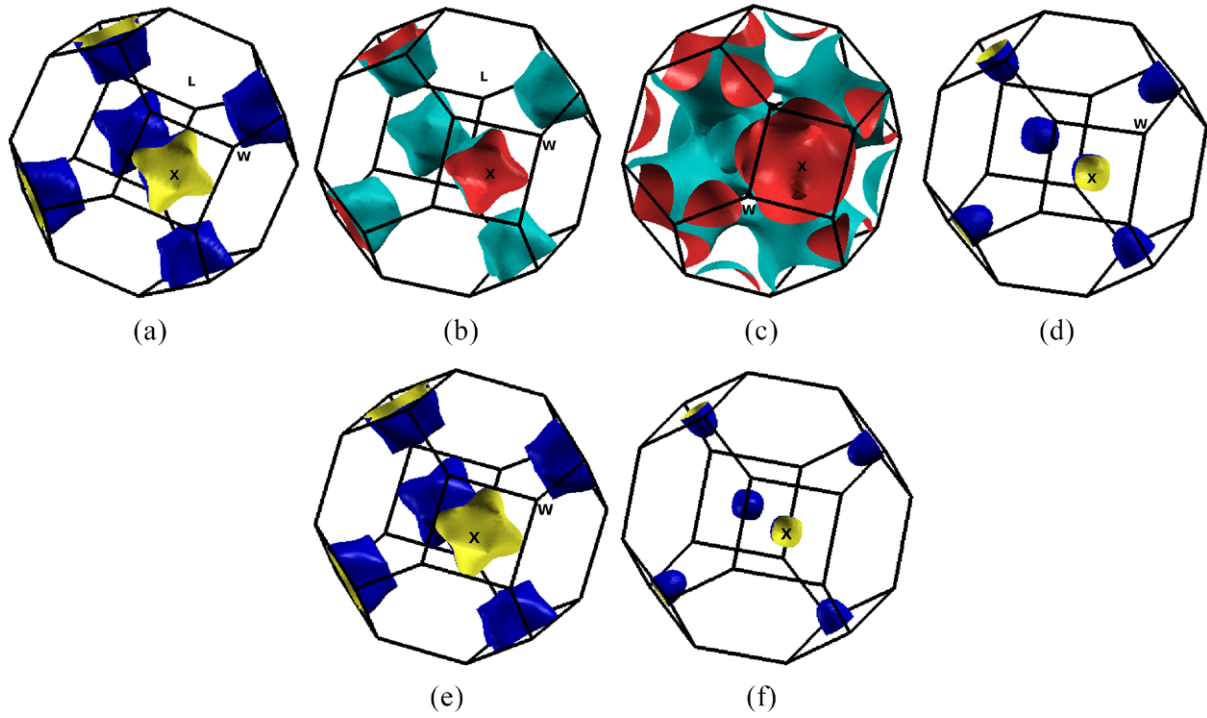
**Figure 3.** Density of states for (a)  $\text{Ni}_2\text{NbAl}$ , (b)  $\text{Ni}_2\text{NbGa}$ , (c)  $\text{Ni}_2\text{NbSn}$  and (d)  $\text{Ni}_2\text{VAl}$ .

the L point. But in the case of present compounds, we have two saddle points in both valence and conduction regions near the  $E_F$ . The flat bands associated with the van Hove singularity at the L-point result in a maximum density of states.

We have also plotted the FS for the bands which are crossing  $E_F$  and are given in figure 4. We have one FS in both  $\text{Ni}_2\text{NbAl}$  and  $\text{Ni}_2\text{NbGa}$  compounds which is of electron nature due to one band crossing the  $E_F$  at X point. In the case of  $\text{Ni}_2\text{NbSn}$  two bands cross  $E_F$ , the blue coloured band (indicated with dotted line) cross the  $E_F$  from conduction band to valence band indicating the electron nature of the band and the remaining red coloured band (indicated with breaking line) cross the  $E_F$  from conduction band to valence band at X point and again from valence to conduction band at W points indicating the mixed character of the band. In a similar way,

we have two FS both having electron nature in  $\text{Ni}_2\text{VAl}$ . In all the compounds we have observed parallel sheets in FS which indicate a nesting feature.

Now we briefly discuss the mechanical properties. The calculated elastic constants satisfy the Born mechanical stability criteria [42] i.e.  $C_{11} > 0$ ,  $C_{44} > 0$ ,  $C_{11} > C_{12}$ , and  $C_{11} + 2C_{12} > 0$  as expected. In table 2, we have given the single crystalline elastic constants along with calculated Debye temperature values. The polycrystalline elastic constants can be calculated from the single crystalline elastic constants the relations between single and polycrystalline elastic constants can be found elsewhere [43–46]. From the same table the calculated Debye temperature ( $\theta_D$ ) values agree well with experiments. BCS theory predicts that  $T_c$  should increase with increasing frequency of the lattice vibrations. For some Heusler



**Figure 4.** Fermi surface for (a)  $\text{Ni}_2\text{NbAl}$ , (b)  $\text{Ni}_2\text{NbGa}$ , ((c), (d))  $\text{Ni}_2\text{NbSn}$  and ((e), (f))  $\text{Ni}_2\text{VAI}$ .

**Table 2.** Calculated single crystalline elastic constants and Debye temperature at ambient condition for  $\text{Ni}_2\text{NbX}$  ( $X = \text{Al, Ga, Sn}$ ) and  $\text{Ni}_2\text{VAI}$ .

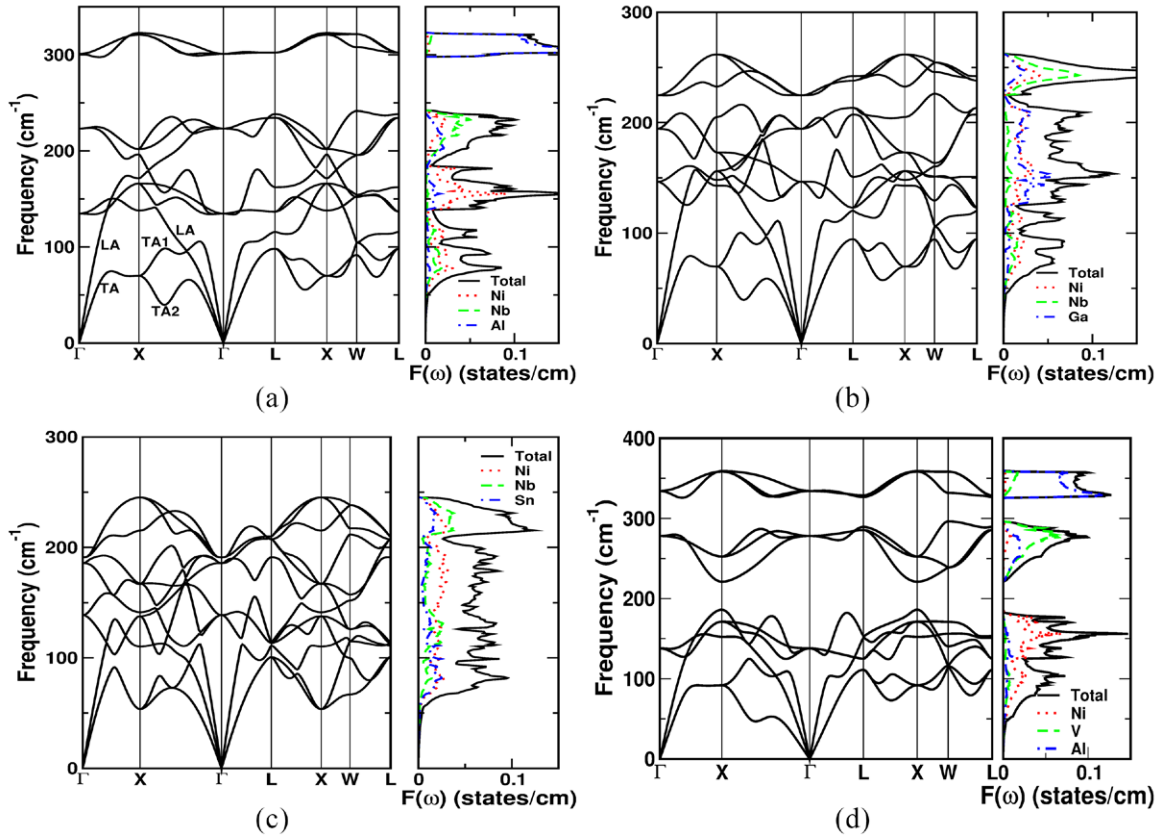
Parameters	$\text{Ni}_2\text{NbAl}$	$\text{Ni}_2\text{NbGa}$	$\text{Ni}_2\text{NbSn}$	$\text{Ni}_2\text{VAI}$
$C_{11}$ (GPa)	212	194	188	200
$C_{12}$ (GPa)	167	176	162	171
$C_{44}$ (GPa)	98	95	72	109
$\Theta_D$ (K)	306	241	219	319
$\Theta_D$ (experimental) (K)	280 [15], 300 [40]	240 [15]	206 [15]	358 [40]

compounds [28]  $T_c$  decreases with the increasing  $\theta_D$ . In the case of  $\text{Ni}_2\text{NbX}$  where  $X = \text{Al, Ga}$ , both  $T_c$  and  $\theta_D$  show the BCS theory behaviour. In the case of  $\text{Ni}_2\text{NbSn}$  it is showing opposite manner. In Pd based Huesler compounds Klimczuk *et al* [28] found a decrease in  $T_c$  with increasing  $\theta_D$ . In their study,  $T_c$  of  $(\text{Sc, Y, Lu})\text{Pd}_2\text{Sn}$  compounds increases with  $N(E_F)$  and found the opposite trend in case of  $\text{APd}_2\text{M}$  ( $A = \text{Zr, Hf}$ ;  $M = \text{Al, In}$ ) compounds. Finally they conclude that the change in  $T_c$  is dependent on the system.

### 3.2. Vibrational and superconducting properties

The calculated phonon dispersion curves are shown in figure 5 along with the total and partial phonon density of states. The primitive unit cells of the present compounds have one formula unit with four atoms which gives 12 phonon branches including three acoustic and nine optical branches. The absence of imaginary phonon frequencies indicates dynamical stability at the ambient condition. The higher frequency optical phonon modes are separated from others in  $\text{Ni}_2\text{NbAl}$ ,  $\text{Ni}_2\text{NbGa}$  and  $\text{Ni}_2\text{VAI}$  compounds. But the same is not found in the case of  $\text{Ni}_2\text{NbSn}$ . This

separation is due to the mass difference between different kinds of atoms in the unit cell. In  $\text{Ni}_2\text{NbAl}$  and  $\text{Ni}_2\text{VAI}$  these higher frequency optical modes are due to the lighter Al atom. In the remaining compounds, it is due to Nb atom. Again in  $\text{Ni}_2\text{NbAl}$  and  $\text{Ni}_2\text{VAI}$ , acoustic and lower frequency optical modes are due to the vibrations of Ni atoms. In all the compounds at the zone centre, we have three optical phonon modes and in that one  $T_{2g}$  mode is Raman active and two  $T_{1u}$  modes are infrared active. Frequency of these  $T_{2g}$  mode in  $\text{Ni}_2\text{NbX}$  ( $X = \text{Al, Ga, Sn}$ ) and  $\text{Ni}_2\text{VAI}$  is 139.8, 148.9, 139.7 and 138.0  $\text{cm}^{-1}$  respectively at the  $\Gamma$  point. In all the compounds we have doubly degenerate acoustic and optical modes from  $\Gamma-X$  and triple degeneracy of same modes is observed along  $X-\Gamma$  direction. This degeneracy is due to the symmetry of the crystal in cubic phase. In all the compounds longitudinal acoustic mode (LA) is interacting with  $T_{2g}$  optical modes. We have observed degenerate transverse acoustic (TA) modes in  $\Gamma-X$  and  $\Gamma-L$  directions in all the compounds. In other directions these TA modes become non degenerate and split into TA1 (high frequency) and TA2 (low frequency) modes. There is an anomaly (dip) in the TA2 mode from the  $X-\Gamma$  direction in all the compounds except



**Figure 5.** Phonon dispersion along with partial phonon density of states for (a)  $\text{Ni}_2\text{NbAl}$ , (b)  $\text{Ni}_2\text{NbGa}$ , (c)  $\text{Ni}_2\text{NbSn}$  and (d)  $\text{Ni}_2\text{VAl}$ .

$\text{Ni}_2\text{NbSn}$ . In the case of  $\text{Ni}_2\text{NbSn}$ , it is observed at the X high symmetry point. As discussed below these dips are related to the FS and reflect the electron phonon coupling.

In metals such dips can arise from FS nesting, i.e. Kohn anomalies in the phonon spectrum. This anomaly is observed in  $\text{Ni}_2\text{MnGa}$  [47–49],  $\text{Ni}_2\text{MnIn}$  [50] and  $\text{Ni}_2\text{MnX}$  ( $X = \text{Sn}, \text{Sb}$ ) [51]. From the phonon dispersion of the present compounds we observed dip (softening) in the lower frequency acoustic mode (TA2) along X– $\Gamma$  direction in both  $\text{Ni}_2\text{NbAl}$ ,  $\text{Ni}_2\text{NbGa}$  and  $\text{Ni}_2\text{VAl}$  compounds and near the X-point in  $\text{Ni}_2\text{NbSn}$ . From the FS, we observed flat portions of X-point FSs in all the investigated compounds to be nested with the similar portion on the other side. FS with nesting vector direction is given in figure 6(a). From this we observed that the nesting vector is of length  $\sim 0.61$  of X–X distance connecting the flat surfaces of FS. In the case of  $\text{Ni}_2\text{NbGa}$  and  $\text{Ni}_2\text{VAl}$  it is at  $\sim 0.64$  and  $\sim 0.67$  of X–X distance. The nesting vector is around  $0.7 \frac{2\pi}{a}$  along the  $\Gamma$ –X direction in  $\text{Ni}_2\text{NbAl}$ . In case of  $\text{Ni}_2\text{NbGa}$  and  $\text{Ni}_2\text{VAl}$  it is observed at  $0.68 \frac{2\pi}{a}$ ,  $0.62 \frac{2\pi}{a}$  respectively in the same direction as  $\text{Ni}_2\text{NbAl}$ . The same nature is observed in  $\text{Ni}_2\text{NbSn}$  at the X high symmetry point. To know the exact ‘q’ vector where the anomaly is observed, we have plotted the acoustic phonon modes from the  $\Gamma$  to X direction for  $\text{Ni}_2\text{NbAl}$ ,  $\text{Ni}_2\text{NbGa}$  and  $\text{Ni}_2\text{VAl}$  compounds as shown in figures 6(b)–(d), where we observed a strong Kohn anomaly at  $\xi = 0.7, 0.68$  and  $0.62$  in  $\text{Ni}_2\text{NbAl}$ ,  $\text{Ni}_2\text{NbGa}$  and  $\text{Ni}_2\text{VAl}$  respectively. The associated phonon branches have mainly Ni

character. In the case of  $\text{Ni}_2\text{MnGa}$ ,  $\text{Ni}_2\text{MnSn}$  and  $\text{Ni}_2\text{MnSb}$  compounds also the same type of atoms are involved in the branches showing Kohn anomalies.

The electron phonon coupling constants ( $\lambda_{\text{ep}}$ ) were extracted from the Eliashberg function ( $\alpha^2F(\omega)$ ) which can be used to determine the  $T_c$  of a conventional phonon mediated superconductor. The calculated  $\alpha^2F(\omega)$  are plotted in figure 7 for all the investigated compounds. From this figure the lower energy phonon modes, which are mainly due to Ni atom at that particular frequency are more involved in the process of scattering the electrons in all the compounds. The  $T_c$  of the present compounds is calculated by using the Allen–Dynes [52] formula which is given by

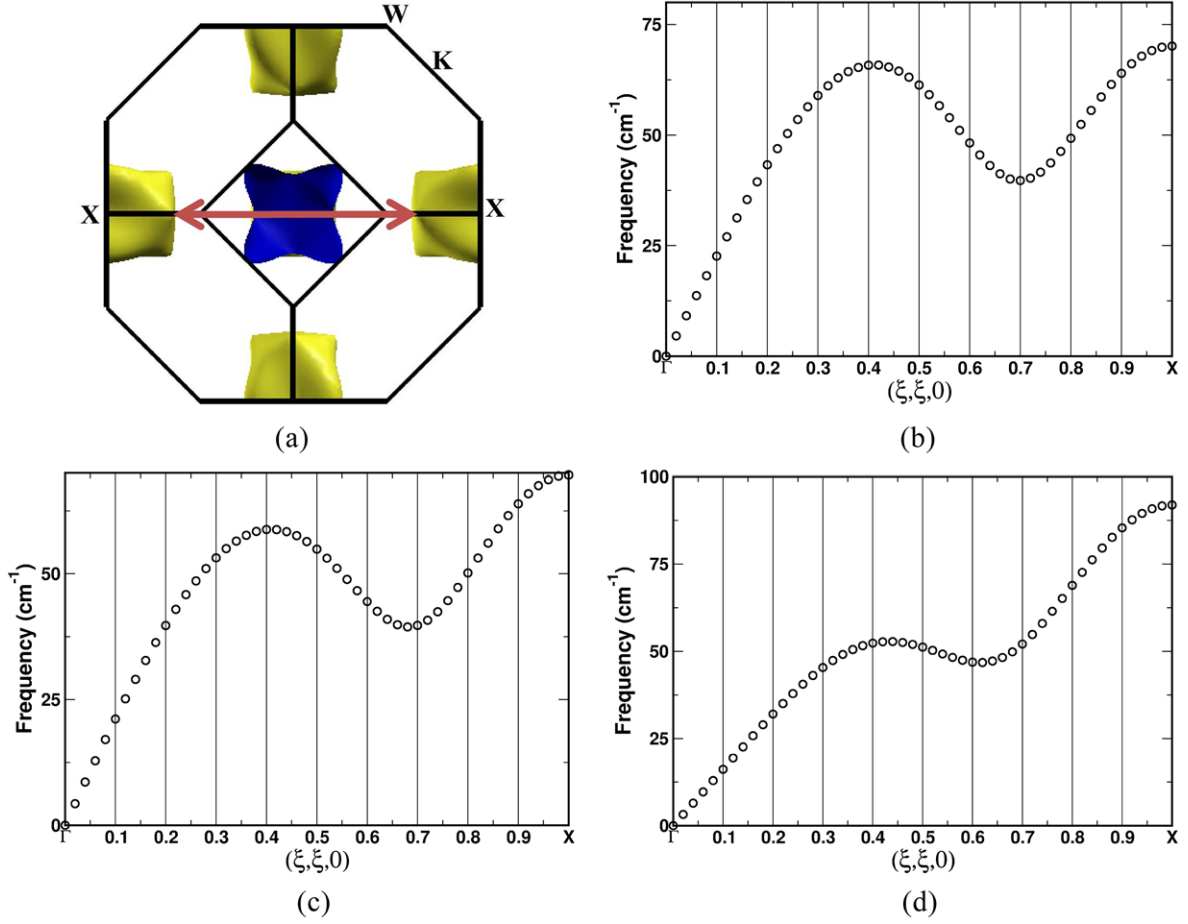
$$T_c = \frac{\omega_{\text{ln}}}{1.2} \exp\left(-\frac{1.04(1 + \lambda_{\text{ep}})}{\lambda_{\text{ep}} - \mu^*(1 + 0.62\lambda_{\text{ep}})}\right) \quad (1)$$

where  $\omega_{\text{ln}}$  is logarithmically averaged phonon frequency,  $\lambda_{\text{ep}}$  is electron phonon coupling constant and  $\mu^*$  is Coulomb pseudopotential which is a parameter that normally takes a value of 0.1–0.15.

The electron–phonon coupling constant ( $\lambda_{\text{ep}}$ ) is usually extracted from the Eliashberg function ( $\alpha^2F(\omega)$ ) which can be used to determine the  $T_c$  of a conventional phonon mediated superconductor.  $\alpha^2F(\omega)$  is

$$\alpha^2F(\omega) = \frac{1}{2\pi N(\epsilon_f)} \sum_{qj} \frac{V_{qj}}{\hbar\omega_{qj}} \delta(\omega - \omega_{qj}) \quad (2)$$





**Figure 6.** (a) Direction of nesting vector and TA2 acoustic phonon mode along the  $\Gamma$ -X direction in (b)  $\text{Ni}_2\text{NbAl}$  (c)  $\text{Ni}_2\text{NbGa}$  and (d)  $\text{Ni}_2\text{VAl}$ .

This function is related to the phonon DOS ( $F(\omega) = \sum_{qj} \delta(\omega - \omega_{qj})$ ) and differs from the phonon DOS by having a weight factor  $1/2\pi N(\epsilon_f)$  inside the summation. In the above formula  $N(\epsilon_f)$  is the electronic density of states at the  $E_F$  and  $\nu_{qj}$  is the phonon line width which can be represented by

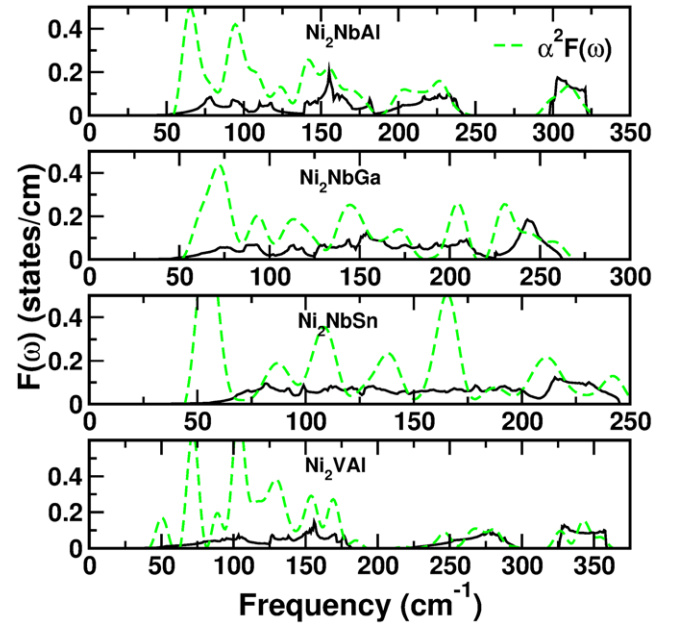
$$\nu_{qj} = 2\pi\omega_{qj} \sum_{km} |g_{(k+q)m, kn}^{qj}|^2 \delta(\epsilon_{kn} - \epsilon_F) \delta(\epsilon_{(k+q)m} - \epsilon_F) \quad (3)$$

where the Dirac delta function expresses the energy conservation conditions and  $g$  is the electron phonon matrix element. The electron phonon coupling constant ( $\lambda_{ep}$ ) can be expressed in terms of  $\alpha^2 F(\omega)$  as

$$\lambda_{ep} = 2 \int \frac{d\omega}{\omega} \alpha^2 F(\omega) = \int \lambda(\omega) d\omega \quad (4)$$

where  $\lambda(\omega) = \frac{2\alpha^2 F(\omega)}{\omega}$

The calculated  $T_c$  values for all the investigated compounds are given in table 3 for  $\mu^*$  values of 0.13 and 0.15. The calculated values agree well with the experiments with  $\mu^* = 0.13$  for  $\text{Ni}_2\text{NbX}$  ( $X = \text{Al}, \text{Ga}, \text{Sn}$ ) compounds. Among the investigated compounds,  $\text{Ni}_2\text{VAl}$  is found to have relatively high  $T_c$  of 3.84 K with high  $\lambda_{ep}$  value 0.68 with  $\mu^* = 0.13$ . As we discussed  $\text{Ni}_2\text{VAl}$  has high electronic DOS and high Sommerfeld coefficient  $\gamma$  at  $E_F$  compared to other compounds indicating the compound to be

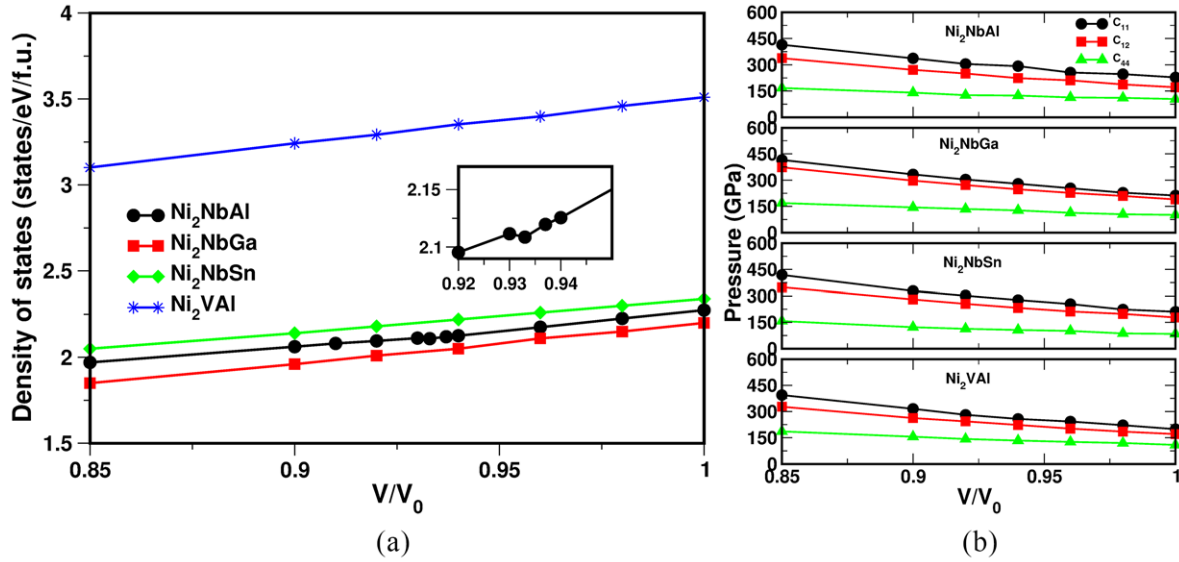


**Figure 7.** Eliasberg function  $\alpha^2 F(\omega)$  and phonon density of states for all compounds.

a superconductor with higher  $T_c$  value than the other compounds. From the calculated  $T_c$  and  $\lambda_{ep}$  values, the superconducting nature of the  $\text{Ni}_2\text{VAl}$  compound is evident.

**Table 3.** Calculated  $T_c$  and  $\lambda_{ep}$  values with experimental reports for  $Ni_2NbX$  ( $X = Al, Ga, Sn$ ) and  $Ni_2VAl$ .

Parameters	$Ni_2NbAl$	$Ni_2NbGa$	$Ni_2NbSn$	$Ni_2VAl$
$T_c$ (experimental)	2.15 [15]	1.54 [15]	2.90 [15], 3.4 [16], 3.4 [17]	—
$T_c$ (this work with $\mu^* = 0.13$ )	1.92	1.18	3.21	3.84
$T_c$ (this work with $\mu^* = 0.15$ )	1.40	0.79	2.60	3.09
$\lambda$ (experiment)	0.52 [15], 0.514 [40]	0.50 [15]	0.61 [15]	—
$\lambda$ (this work)	0.56	0.50	0.68	0.68

**Figure 8.** (a) Electronic density of states under compression and (b) elastic constants under compression for all compounds.

These calculated values are in the range of the present  $Ni_2NbX$  ( $X = Al, Ga, Sn$ ) and other Ni based superconducting Heusler compound  $ZrNi_2Ga$  [31].

### 3.3. Under compression

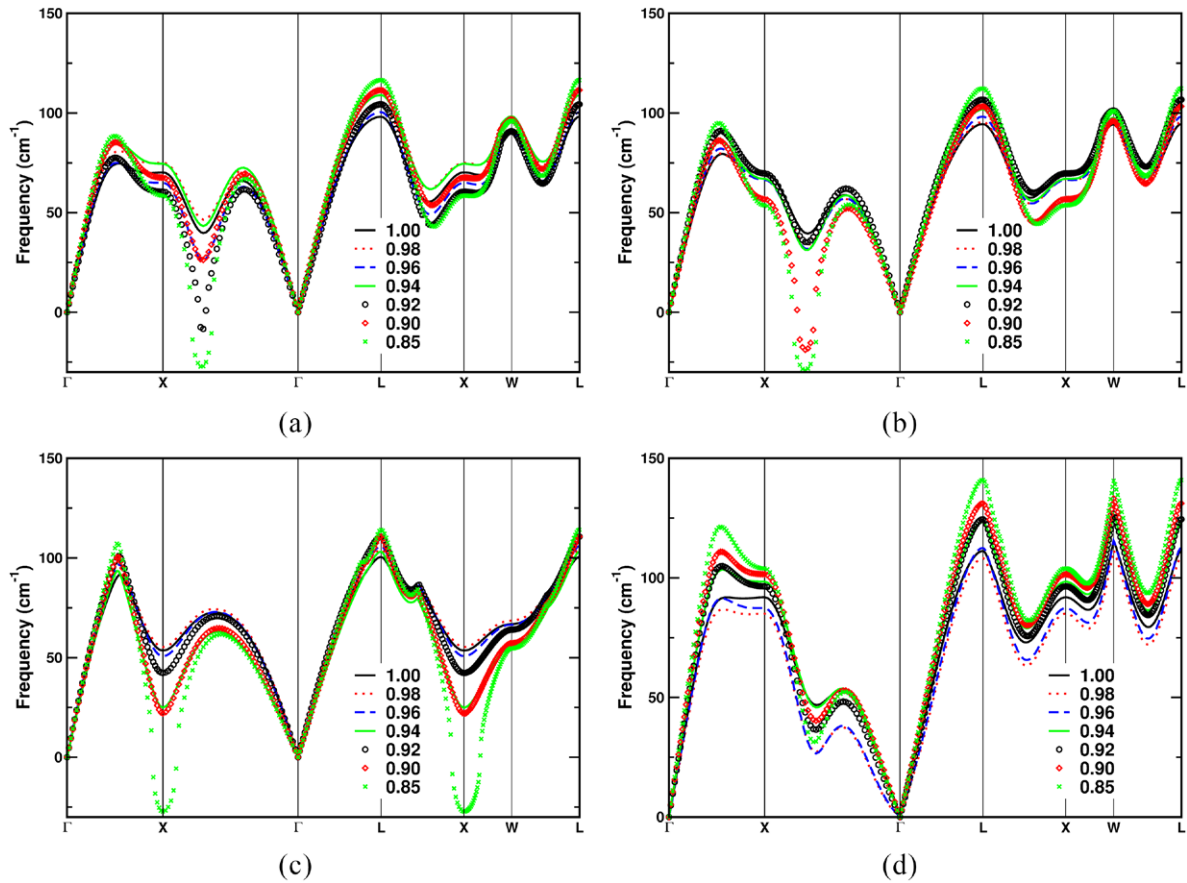
As mentioned, pressure is a clean tuning parameter for exploring the relationships between superconductivity and other physical properties. This motivated us to proceed further to see the effect of pressure on the above mentioned properties for the compounds studied. For all the compounds we compressed the volume up to  $-15\%$  of the initial volume. The pressure effect on the band structure and FS of  $Ni_2NbAl$  compound is already reported in our previous work [20], where we find an extra band to cross the  $E_F$  at the compression of  $V/V_0 = 0.93$  (pressure of 17 GPa) at the X point and due to this we have an extra FS at the same point having the electron character. In the case of remaining two compounds, we have not observed any change in band structure and FS topology under compression as mentioned above.

The electronic DOS under compression and the values are shown in figure 8(a). From the plot, the total electronic DOS linearly decreases with pressure in  $Ni_2NbGa$ ,  $Ni_2NbSn$  and  $Ni_2VAl$  compounds but in the case of  $Ni_2NbAl$  it is non-linear at  $V/V_0 = 0.93$  and is clearly represented in the inset.

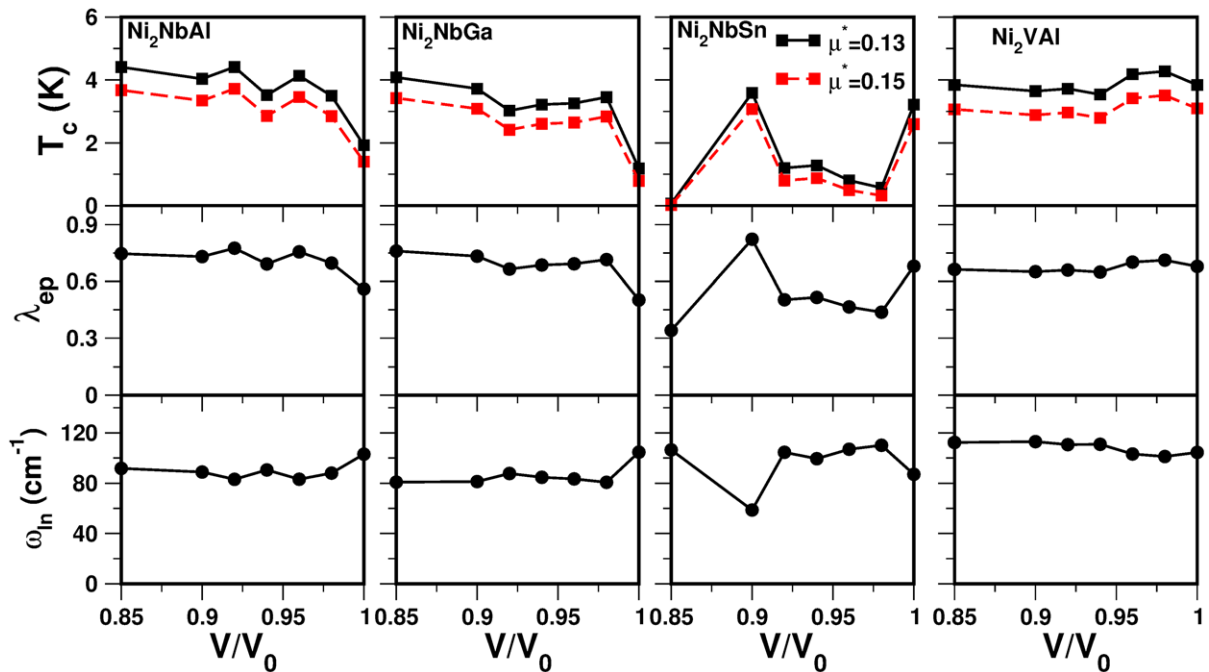
We have also calculated the single crystalline elastic constants for all the compounds under compression to check the effect of pressure on the mechanical stability in the present

compounds and are plotted in figure 8(b). From this we can observe that all the compounds are satisfying Born's [42] stability criteria under compression indicating the mechanically stable nature of the present compounds under the compression range we have studied. We also observed that the values of three independent elastic constants increases under compression in all the compounds as usual. It is also observed that  $C_{11}$ ,  $C_{12}$  are more sensitive to pressure while  $C_{44}$  is quite insensitive to pressure for the same compounds, where  $C_{44}$  is related to transverse distortion which is almost flat indicating the effect of pressure on this to be weaker.

We calculated the phonon dispersion under compression. We find hardening of frequencies for all the modes except in the lowest frequency acoustic mode which softens under compression in all the compounds. In figure 9, we have given only the lower frequency acoustic mode under compression for all the compounds to show the softening nature in that particular mode. From this figure, we have observed that the softening becomes more pronounced under compression at the same 'q' vector where the Kohn anomaly is found in all compounds. The softening in phonon frequency, corresponding to the Kohn anomaly, under pressure in the Nb compounds implies a pressure dependent structural phase transition. Near the transition pressure this is presumably of charge density wave character. In  $Ni_2NbAl$  it is observed to be imaginary at compression  $V/V_0 = 0.92$  and  $0.85$ . In this compound, we find an extra FS at  $V/V_0 = 0.93$ . The change in the FS topology could be reason for the imaginary mode at  $V/V_0 = 0.92$  compression.



**Figure 9.** Softening of the lower frequency acoustic mode under compression in (a)  $\text{Ni}_2\text{NbAl}$ , (b)  $\text{Ni}_2\text{NbGa}$ , (c)  $\text{Ni}_2\text{NbSn}$  and (d)  $\text{Ni}_2\text{VAl}$ .



**Figure 10.** Logarithmic frequency, electron–phonon coupling and superconducting transition temperature under compression for  $\text{Ni}_2\text{NbX}$  ( $X = \text{Al}, \text{Ga}$  and  $\text{Sn}$ ) and  $\text{Ni}_2\text{VAl}$  compounds.

And also the nesting feature under compression becomes more prominent due to the increase in the size of the FS under compression. In the remaining compounds, no extra FS is observed but the size of the FS increases under compression

which might lead to an increase in the effect of nesting under compression. In  $\text{Ni}_2\text{MnSb}$  the authors of [51] observed the imaginary frequency in the TA2 mode at ambient condition which is due to the presence of the Kohn anomaly in this

system. The imaginary frequencies under compression in the TA2 mode in all the compounds may be due to the presence of the same Kohn anomaly under compression. This acoustic mode softening is also observed in other Heusler compounds. In HfPd<sub>2</sub>Al [53] it is observed at the pressure of 7.5 GPa, Pd<sub>2</sub>ZrAl [32] at ambient condition. In the case of YPd<sub>2</sub>Sn the authors of [54] found an anomaly in the transverse acoustic mode and reported that to be the reason for the increase in the electron–phonon coupling parameter of that phonon branch. The anomaly in the transverse acoustic mode is also observed in non Heusler compounds [55–59], where the authors reported that these phonon anomalies play an important role in understanding superconductivity in those compounds. As discussed previously, we have observed the anomalies in the phonon frequencies at the X–Γ point in Ni<sub>2</sub>NbAl, Ni<sub>2</sub>NbGa and Ni<sub>2</sub>VAl compounds and at the X point in Ni<sub>2</sub>NbSn. We already know that the lower frequency phonon modes would contribute more to the electron phonon coupling, which further has an impact on the  $T_c$  of that material. This indicate that the softening may lead to change in the  $\lambda_{ep}$  and  $T_c$  in these compounds.

We have calculated the electron–phonon coupling constant and superconducting transition temperature for all the compounds under compression and these are shown in figure 10. From these plots, all the compounds show a non-monotonic variation in  $T_c$  and the electron–phonon coupling constant under compression behaves in the opposite manner to  $\omega_{in}$ . In prior work [60, 61] it was reported that the softening of the phonon DOS leads to the increase in the  $T_c$  of that material. In Ni<sub>2</sub>NbAl, the  $T_c$  plot under compression in our previous work [20] is different from this work. In our previous work we used ultrasoft pseudopotentials and also we did not find any softening nature in the phonon dispersion curve as well as phonon density of states. In the present work we have used norm-conserving pseudopotentials. These are more difficult to converge and require more computational resources but are more reliable.

#### 4. Conclusions

We have studied the ground state, electronic structure, mechanical, vibrational and superconducting properties of Ni<sub>2</sub>NbX (X = Al, Ga and Sn) and Ni<sub>2</sub>VAl Heusler compounds using density functional theory within a generalized gradient approximation at ambient and under compression. Our calculated ground state properties agree well with experimental and other theoretical results. From these calculations, at the  $E_F$  the contribution in DOS is mainly from Ni  $d_{eg}$  states with the admixture of Nb/V  $d_{t_{2g}}$  states in all the compounds. The FS resulting from the bands crossing the  $E_F$  are also shown and observed that one electron pocket exist in both Ni<sub>2</sub>NbAl and Ni<sub>2</sub>NbGa compounds and two FSs of electron nature are found in the case of Ni<sub>2</sub>NbSn and Ni<sub>2</sub>VAl compounds. Nesting features in FS are evident in all the compounds. The calculated single crystalline elastic constants satisfy the Born stability criteria indicating the mechanical stability of the investigated compounds both at ambient and under compression. The absence of imaginary frequencies in phonon dispersion calculations indicates the dynamical stability of the

present compounds. Kohn anomaly is observed in TA2 mode in all the compounds which might be due to the nesting in the FS. The computed  $T_c$  using the Allen–Dynes formula agrees well with the experimental results. Importantly, Ni<sub>2</sub>VAl is predicted to be a superconductor which have  $T_c$  value in the range of Heusler compounds. From the calculated Eliashberg function ( $\alpha^2F(\omega)$ ) we have observed that Ni atom contribution is more towards  $T_c$  in all the compounds. Under compression we have observed the change in the band structure of Ni<sub>2</sub>NbAl compound where we find an extra band to cross the  $E_F$ . In addition we also observed a pronounced softening in the Kohn anomalies existing in TA2 mode under compression. This phonon softening again lead to softening in the phonon density of states. A non-monotonic variation in the  $T_c$  under compression is noted which is due to softening in the lower frequency acoustic mode and phonon DOS. Among the compounds studied, we observe that the variation in  $T_c$  under pressure is minimal in Ni<sub>2</sub>VAl. It will be of interest to experimentally search for superconductivity in Ni<sub>2</sub>VAl and study its pressure dependence.

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#### References

- [1] de Groot R A, Mueller F M, van Engen P G and Buschow K H J 1983 *Phys. Rev. Lett.* **50** 2024–7
- [2] Kandpal H C, Fecher G H and Felser C 2007 *J. Phys. D: Appl. Phys.* **40** 1507–23
- [3] Galehgirian S and Ahmadian F 2015 *Solid State Commun.* **202** 52–7
- [4] Lei F, Tang C, Wang S and He W 2011 *J. Alloy. Compd.* **509** 5187–9
- [5] Gofryk K, Kaczorowski D, Plackowski T, Leithe-Jasper A and Yu G 2005 *Phys. Rev. B* **72** 094409
- [6] Kaczorowski D, Gofryk K, Plackowski T, Leithe-Jasper A and Yu G 2005 *J. Magn. Magn. Mater.* **290–1** 573–9
- [7] Gofryk K, Kaczorowski D and Czopnik A 2005 *Solid State Commun.* **133** 625–8
- [8] Nakamura H, Kitaoka Y, Asayama K, Ōnuki Y and Komatsubara T 1988 *J. Magn. Magn. Mater.* **76–7** 467–8
- [9] Takayanagi S, Woods S B, Wada N, Watanabe T, Ōnuki Y, Kobori A, Komatsubara T, Imai M and Asano H 1988 *J. Magn. Magn. Mater.* **76–7** 281–2
- [10] Wuttig M, Li J and Craciunescu C 2001 *Scr. Mater.* **44** 2393–7
- [11] Sutou Y, Imano Y, Koeda N, Omori T, Kainuma R, Ishida K and Oikawa K 2004 *Appl. Phys. Lett.* **85** 4358–60
- [12] Tobola J and Pierre J 2000 *J. Alloy. Compd.* **296** 243–52
- [13] Bhattacharya S, Pope A L, Littleton I V, Tritt T M, Ponnambalam V, Xia Y and Poon S J 2000 *Appl. Phys. Lett.* **77** 2476

- [14] Ishikawa M, Jorda J-L and Junod A 1982 *Superconductivity in d- and f-band Metals* ed W Buckel and W Weber (Karlsruhe: Kernforschungszentrum) p 141
- [15] Waki S, Yamaguchi Y and Mitsugi K 1985 *J. Phys. Soc. Japan* **54** 1673–6
- [16] Boff M A S, Fraga G L F, Brandão D E and Gomes A A 1996 *J. Magn. Magn. Mater.* **153** 135–40
- [17] Wernick J H, Hull G W, Geballe T H, Bernardini J E and Waszczak J V 1983 *Mater. Lett.* **2** 90–2
- [18] Benndorf C, Niehaus O, Eckert H and Janka O 2015 *Z. Anorg. Allg. Chem.* **641** 168–75
- [19] Lin W and Freeman A J 1992 *Phys. Rev. B* **45** 61
- [20] Sreenivasa Reddy P V and Kanchana V 2014 *J. Alloy. Compd.* **616** 527–34
- [21] Matthias B T 1953 *Phys. Rev.* **92** 874
- [22] Matthias B T 1955 *Phys. Rev.* **97** 74
- [23] Ishizuka M, Iketani M and Endo S 2000 *Phys. Rev. B* **61** R3823
- [24] Vaitheeswaran G, Shameen Banu I B and Rajagopalan M 2000 *Solid State Commun.* **116** 401–4
- [25] Suzuki N and Otani M 2002 *J. Phys.: Condens. Matter* **14** 10869
- [26] Suzuki N and Otani M 2007 *J. Phys.: Condens. Matter* **19** 125206
- [27] Berk N F and Schrieffer J R 1966 *Phys. Rev. Lett.* **17** 433
- [28] Klimczuk T et al 2012 *Phys. Rev. B* **85** 174505
- [29] Singh P P 2006 *Phys. Rev. Lett.* **97** 247002
- [30] van Hove L 1953 *Phys. Rev.* **89** 1189
- [31] Ming W, Liu Y, Zhang W, Zhao J and Yao Y 2009 *J. Phys.: Condens. Matter* **21** 075501
- [32] Winterlik J, Fecher G H, Thomas A and Felser C 2009 *Phys. Rev. B* **79** 064508
- [33] Ramesh Kumar K, Chunchu V and Thamizhavel A 2013 *J. Appl. Phys.* **113** 17E155
- [34] Blaha P, Schwarz K, Sorantin P and Tricky S B 1990 *Comput. Phys. Commun.* **59** 399–415
- [35] Perdew J P, Burke K and Ernzerhof M 1996 *Phys. Rev. Lett.* **77** 3865
- [36] Monkhorst H J and Pack J D 1976 *Phys. Rev. B* **13** 5188
- [37] Blöchl P, Jepsen O and Andersen O K 1994 *Phys. Rev. B* **49** 16223
- [38] Birch F 1947 *Phys. Rev.* **71** 809
- [39] Giannozzi P et al 2009 *J. Phys.: Condens. Matter* **21** 395502
- [40] da Rocha F S, Fraga G L F, Brandão D E, da Silva C M and Gomes A A 1999 *Physica B* **269** 154–62
- [41] Villars P and Calvert L D 1986 *Pearson's Handbook of Crystallographic Data for Intermetallic Phases* (Metals Park, OH: American Society for Metals)
- [42] Born M 1939 *J. Chem. Phys.* **7** 591
- [43] Kanchana V 2009 *Eur. Phys. Lett.* **87** 26006
- [44] Kanchana V, Vaitheeswaran G, Yanming M, Yu X, Svane A and Eriksson O 2009 *Phys. Rev. B* **80** 125108
- [45] Kanchana V, Vaitheeswaran G, Svane A and Delin A 2006 *J. Phys.: Condens. Matter* **18** 9615
- [46] Kanchana V, Vaitheeswaran G and Svane A 2008 *J. Alloy. Compd.* **455** 480
- [47] Bungaro C, Rabe K M and Dal Corso A 2003 *Phys. Rev. B* **68** 134104
- [48] Zayak A T and Entel P 2003 *Phys. Rev. B* **68** 132402
- [49] Ayuela A, Enkovaara J, Ullakko K and Nieminen R M 1999 *J. Phys.: Condens. Matter* **11** 2017–26
- [50] Ağduk S and Gökoğlu G 2011 *Euro. Phys. J. B* **79** 509–14
- [51] Ağduk S and Gökoğlu G 2012 *J. Alloy. Compd.* **511** 9–13
- [52] Allen P B and Dynes R C 1987 *J. Phys. C: Solid State Phys.* **8** L158
- [53] Wiendlocha B, Winiarski M J, Muras M, Zvoriste-Walters C, Griveau J-C, Heathman S, Gazda M and Klimczuk T 2015 *Phys. Rev. B* **91** 024509
- [54] Tütüncü H M and Srivastava G P 2014 *J. Appl. Phys.* **116** 013907
- [55] Isaeva E I, Ahuja R, Simak S I, Lichtenstein A I, Vekilov Y K, Johansson B and Abrikosov I A 2005 *Phys. Rev. B* **72** 064515
- [56] Isaev E I, Simak S I, Abrikosov I A, Ahuja R, Vekilov Y K, Katsnelson M I, Lichtenstein A I and Johansson B 2007 *J. Appl. Phys.* **101** 123519
- [57] Bağcı S, Duman S, Tütüncü H M and Srivastava G P 2008 *Phys. Rev. B* **78** 174504
- [58] Bağcı S, Tütüncü H M, Duman S and Srivastava G P 2010 *Phys. Rev. B* **81** 144507
- [59] Tütüncü H M, Bağcı S, Srivastava G P and Akbulut A 2012 *J. Phys.: Condens. Matter* **24** 455704
- [60] Allen P B and Cohen M L 1972 *Phys. Rev. Lett.* **29** 1593
- [61] Wühl H, Eichler A and Wittig J 1973 *Phys. Rev. Lett.* **31** 1393